
Material Modeling of TWIP-Steels: Applications to Sheet Metal Forming Simulations

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Salzgitter Mannesmann Forschung GmbH

Swerea KIMAB AB

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Material Modeling of TWIP-Steels: Applications to Sheet Metal Forming Simulations

Outline

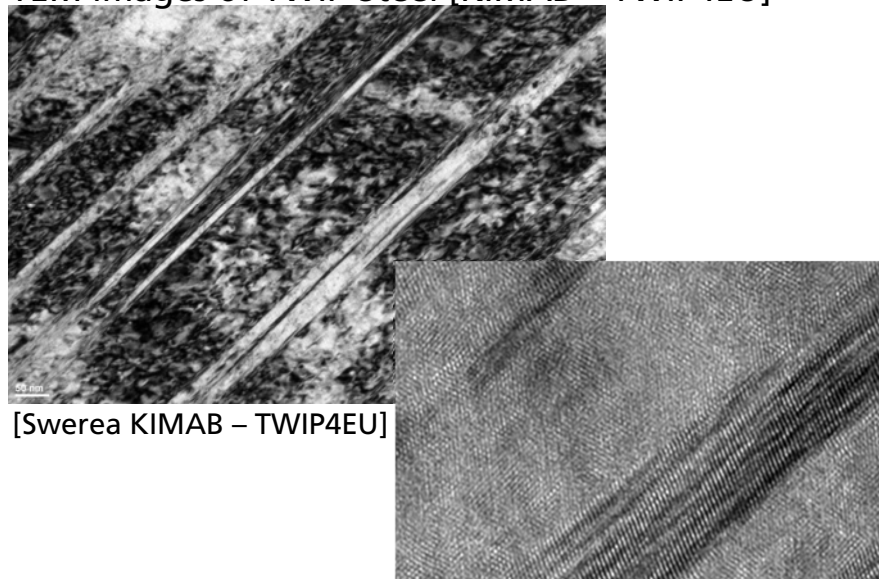
- Introduction and motivation
- Experimental results for TWIP steel
- Macroscopic constitutive model for TWIP steel
- Numerical simulations
- Conclusions and outlook

Introduction and motivation

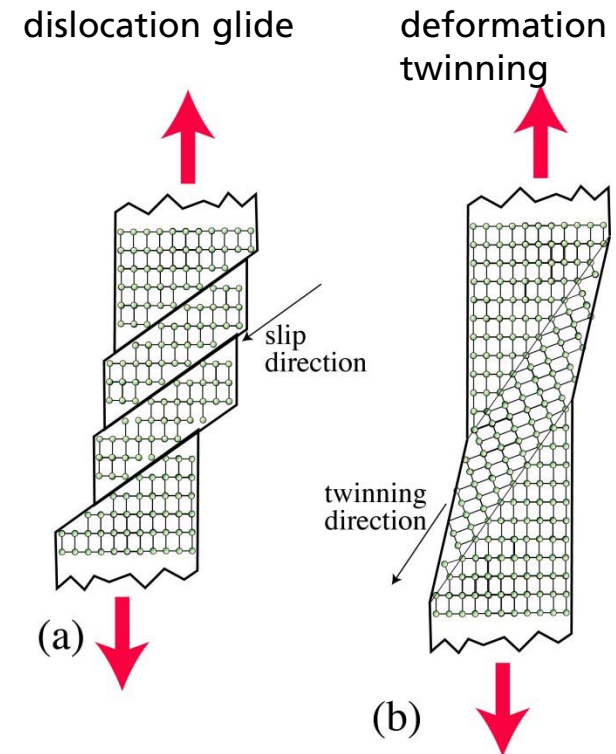
TWIP steels

- TWIP-steel (**t**winning induced **p**lasticity)
 - Class of high manganese austenitic steels
 - Deformation mechanism involves dislocation glide as well as twinning
 - Dynamic Hall-Petch effect leads to high hardening rates

TEM images of TWIP-Steel [KIMAB – TWIP4EU]

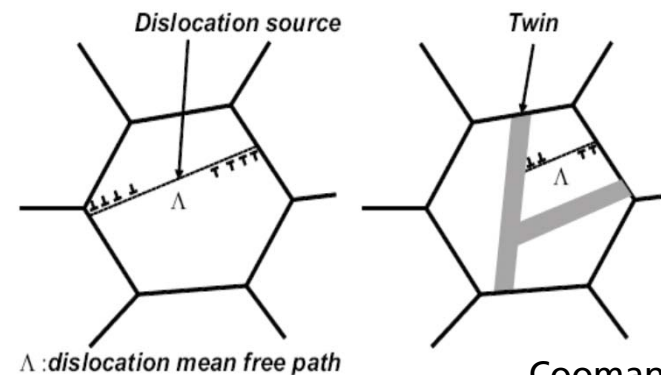


[Swerea KIMAB – TWIP4EU]



<http://www.doitpoms.ac.uk>

Dynamic Hall-Petch effect

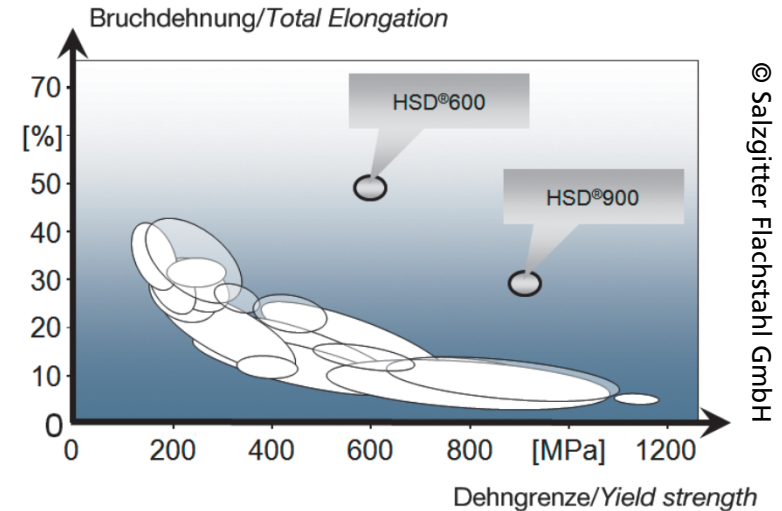
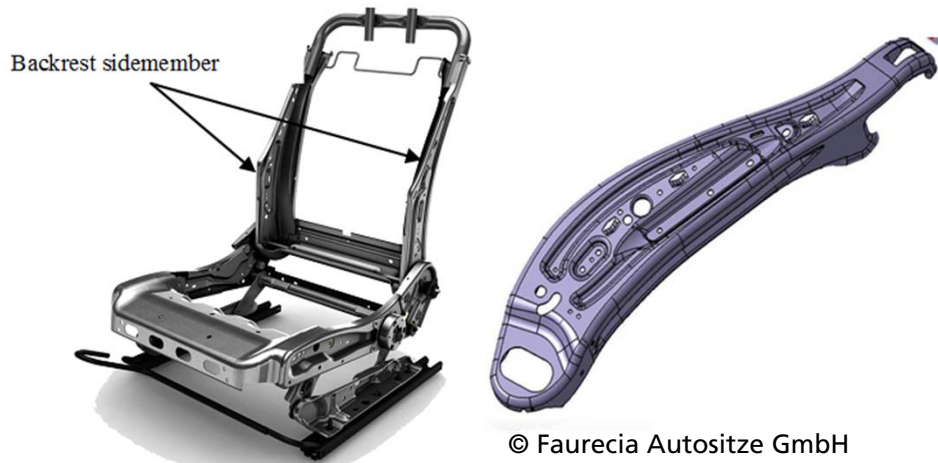


Cooman *et al.*, 2012

Introduction and motivation

TWIP steels

- Superior material properties of TWIP-steels
 - High strength
 - High ductility
 - Lower density/weight
- Typical applications
 - Light weight construction
 - Crash relevant components



HSD®-steel torsion specimen
© Salzgitter Flachstahl GmbH



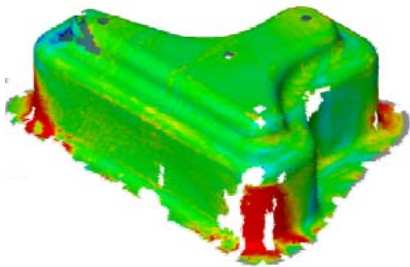
Crush testing under the same load with the same sheet thickness

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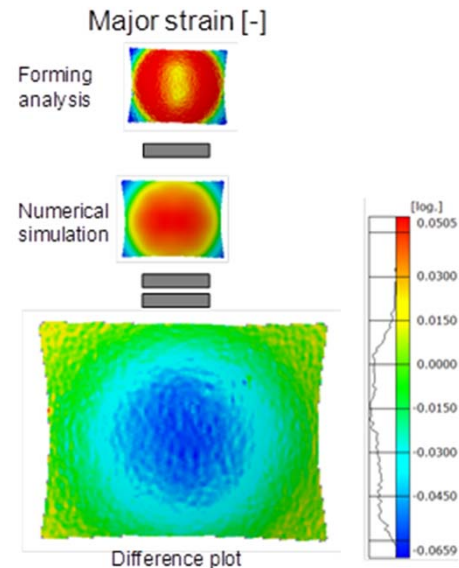
Introduction and motivation

TWIP steels

- Challenges to introduce TWIP steels on the automotive market:
 - Different material behavior compared to conventional steels
 - Deviations between experiments and simulations if standard material models are used to describe the behavior of TWIP steels.



Preliminary study from SZMF: Difference between the strain field obtained from experimental data and the simulation using standard material models (red: deviation of approx. 10%)



Nakajima-test was used to evaluate standard material model: Up to 6 percentage points deviation from experiments (Salzgitter Mannesmann Forschung SZMF)

- Appropriate material models for TWIP steels should be available to introduce the material on the market.

Outline

- Introduction and motivation
- **Analysis of the experimental results for TWIP steel**
- Macroscopic constitutive model for TWIP steel
- Numerical simulations
- Conclusions and outlook

Experimental analysis

Material characterization

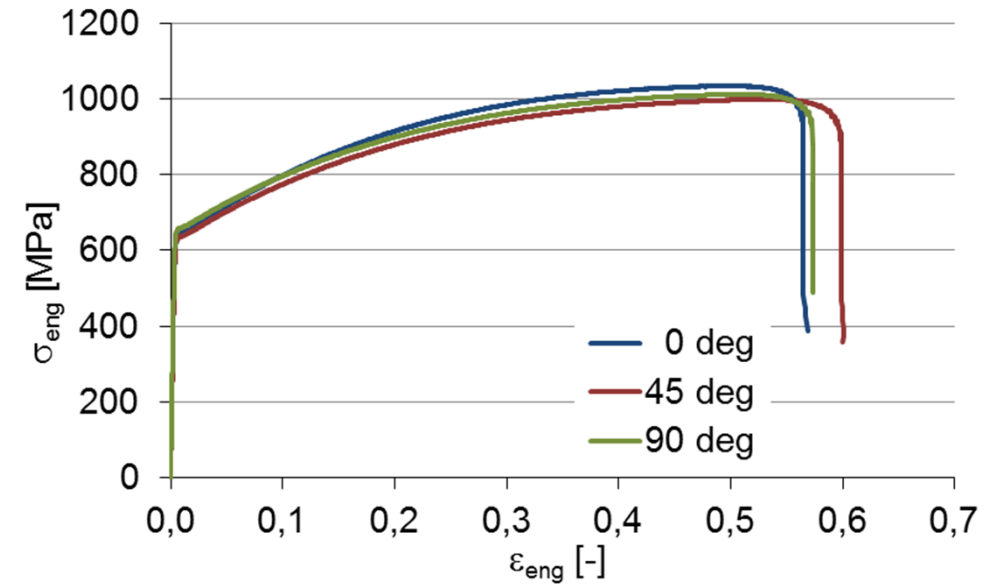
- TWIP-steel material provided by Salzgitter AG
- TWIP steel alloy Fe-15Mn-0.7C-2.5Al-2.5Si
→ Alloying concept: No delayed fracture
- Sheet thickness = 1.5 mm

Mechanical characterization was aimed at the following objectives:

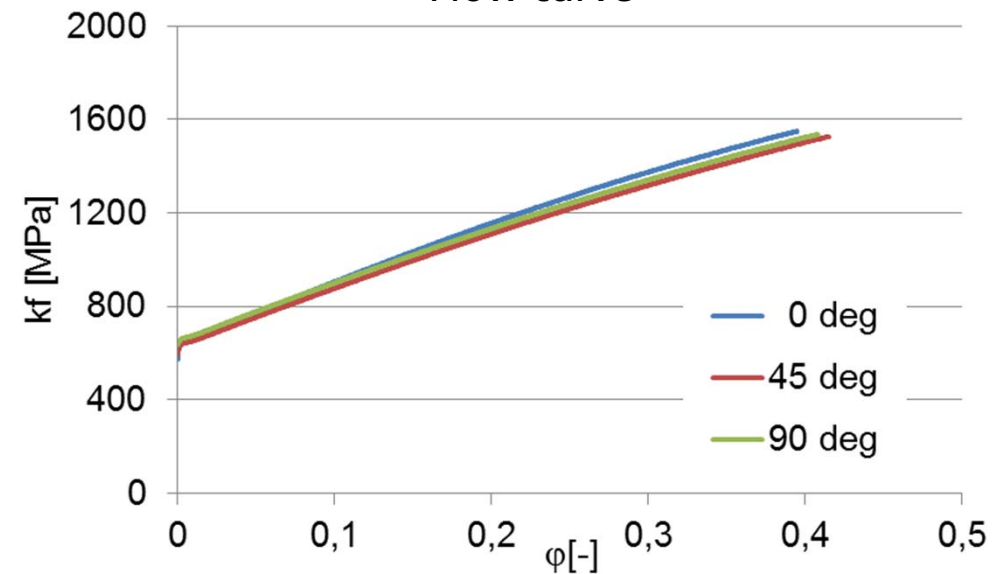
- Mechanical quantities used in sheet metal forming
- Spring-back behavior
- Formability of the material

Specific values	0 deg.	45 deg.	90 deg.
$R_{p0.20}$ [MPa]	633	630	652
Tensile strength [MPa]	1035	999	1012
Ultimate strain A_{25} [-]	56,7	59,9	57,1
r-value (incr.) [-]	0,79	0,96	1,00
Young's modulus [MPa]	172952	168000	175000

Stress-strain curve



Flow curve

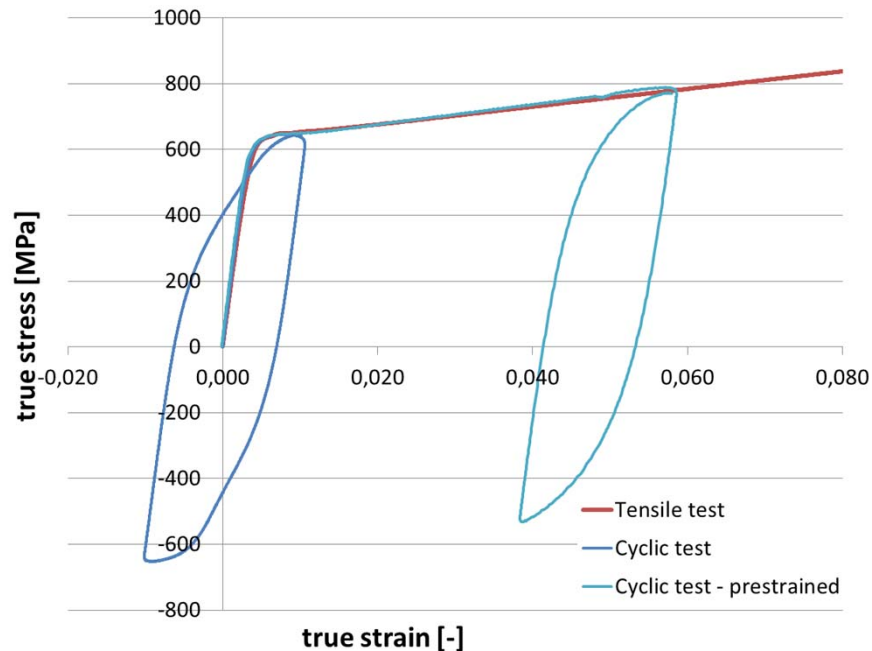


Experimental analysis

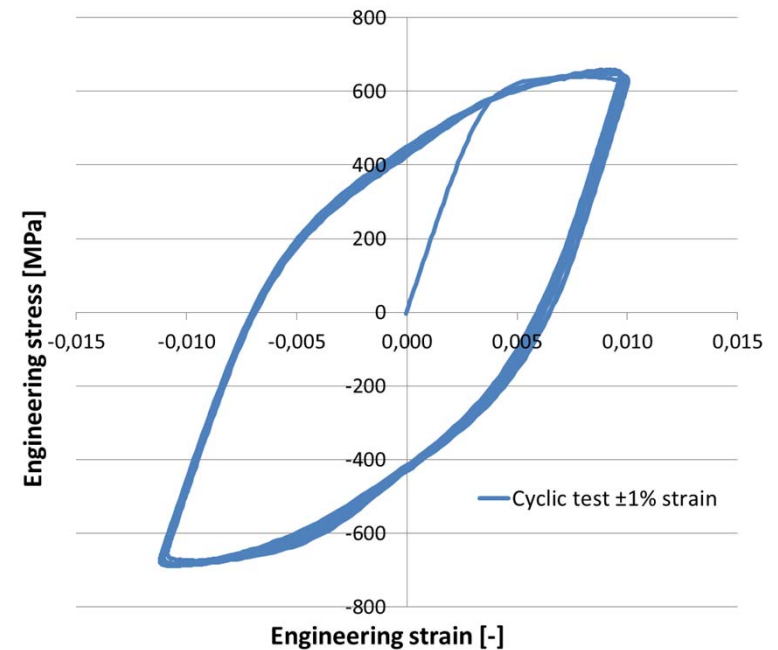
Cyclic behavior

- Significant Bauschinger effect is revealed
- No significant isotropic hardening: after several load cycles no substantial expansion of the hysteresis can be observed
- Strain amplitude limited due to buckling of the specimen

Cyclic loading and cyclic loading with pre-strain



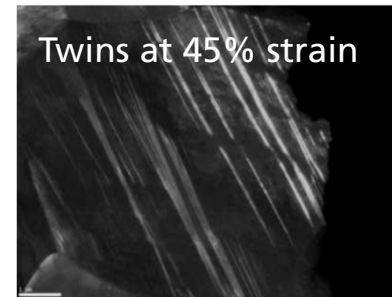
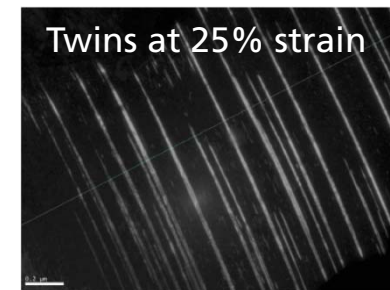
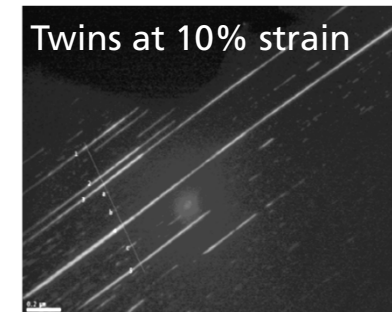
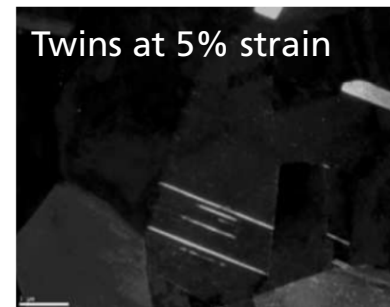
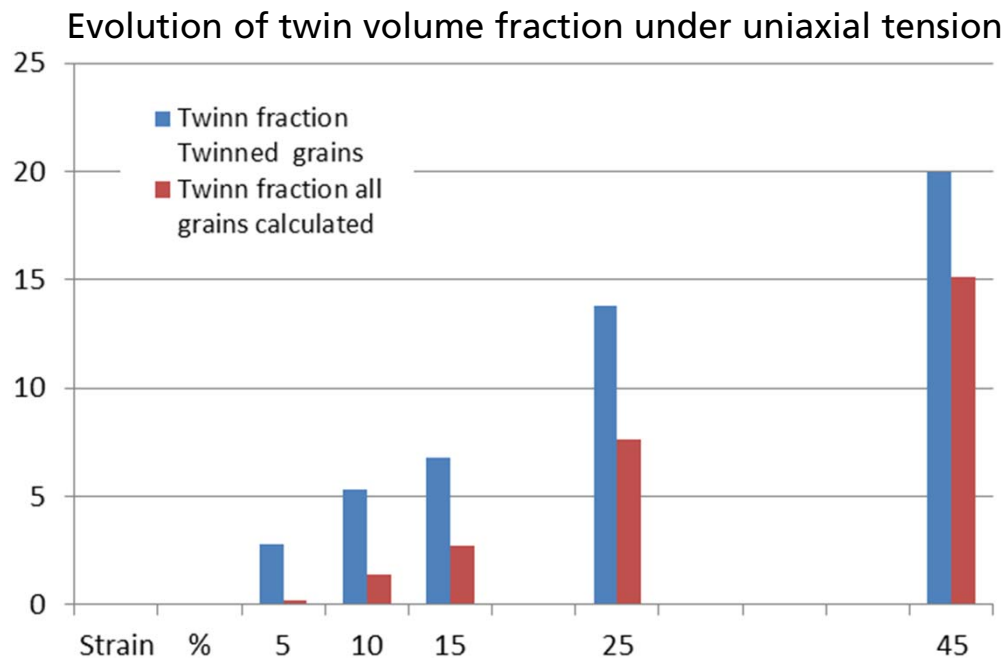
Cyclic loading – several load cycles



Experimental analysis

Microstructure

- **TEM-analysis** to evaluate twin volume fraction and dislocation density
- Twinning evolution during uniaxial tensile loading:
 - Twin volume fraction increases with increasing strain level
 - Practically no twins were observed for strains $< 5\%$
- Input for parameter identification of the macroscopic model



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Macroscopic constitutive modelling of TWIP steels

3D-Elasto-plastic model for TWIP-Steels

Aim:

- Macroscopic elasto-plastic material model for TWIP-steels
- Physically motivated model based on [Bouaziz et al. 2011]
 - Twin volume fraction
 - Dislocation density } as internal variables

TWIP4EU - material model:

- Current extensions of the base model:
 - Extension from 1D-formulation to a 3D-formulation
 - Stress-dependent twinning evolution
 - Armstrong-Frederick type approach for the evolution of inner variables
 - Anisotropic yield function

Macroscopic constitutive modelling of TWIP steels

3D-Elasto-plastic model for TWIP-Steels

Yield function: $f = \|\boldsymbol{\sigma}^D - \mathbf{X}^D\| - \sqrt{\frac{2}{3}} (\sigma_0 + \sigma_f)$ (isotropic formulation)

- Isotropic hardening: $\sigma_f = \alpha M \mu b \sqrt{\rho}$
- Kinematic hardening: $\mathbf{X} = M \frac{\mu b}{L} \mathbf{n}$

Constants

α : constant
 M : the average Taylor factor
 μ : the shear modulus,
 b : the Burgers vector

Internal variables:

ρ : statistical stored dislocation density
 L : geometrical length scale of the microstructure
 \mathbf{n} : number of dislocations stopped at the boundary
 F : twin volume fraction

Macroscopic constitutive modelling of TWIP steels

3D-Elasto-plastic model for TWIP-Steels

Evolution equations for inner variables

- Statistical stored dislocation density

$$\frac{d\rho}{ds^p} = M \left(\frac{1 - J(\mathbf{n})/n_0}{bL} + \frac{k}{b} \sqrt{\rho} - f\rho \right)$$

s^p : equivalent plastic strain

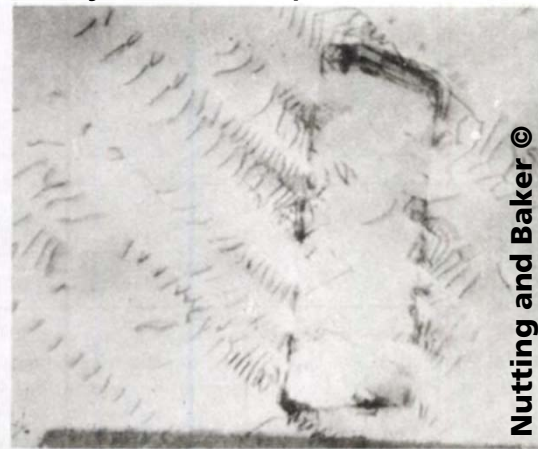
- The number of dislocations which have been stopped at the boundary per slip band

$$\frac{dn}{ds^p} = \frac{\lambda}{b} \left(\frac{2}{3} s^p - \dot{s}^p \frac{\mathbf{n}}{n_0} \right)$$

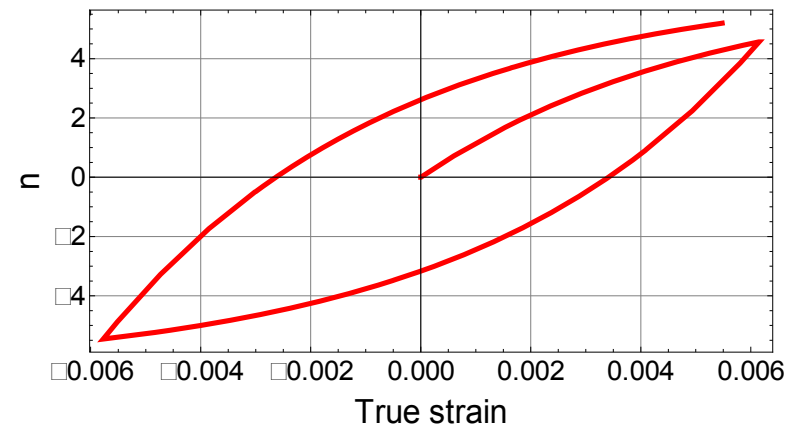
λ : mean spacing between slip bands,

n_0 : maximum number of dislocation loops at the boundary

Physical interpretation



- Assumption: Reversible nature of the dislocation flux under cyclic loading



Macroscopic constitutive modelling of TWIP steels

3D-Elasto-plastic model for TWIP-Steels

Evolution equations for inner variables

■ Statistical stored dislocation density

$$\frac{d\rho}{ds^p} = M \left(\frac{1 - J(\mathbf{n})/n_0}{bL} + \frac{k}{b} \sqrt{\rho} - f\rho \right) \quad s^p: \text{equivalent plastic strain}$$

■ The number of dislocations which have been stopped at the boundary per slip band

$$\frac{dn}{ds^p} = \frac{\lambda}{b} \left(\frac{2}{3} s^p - \dot{s}^p \frac{\mathbf{n}}{n_0} \right) \quad \begin{array}{l} \lambda: \text{mean spacing between slip bands,} \\ n_0: \text{maximum number of dislocation loops at the boundary} \end{array}$$

■ Twin volume fraction

$$F = F_0 \left(1 - e^{-\beta(s^p - \varepsilon_{init})} \right)^m \quad m: \text{the stacking fault energy parameter}$$

■ Geometrical length scale of the microstructure

$$\frac{1}{L} = \frac{1}{d} + \frac{1}{t} \quad \text{where} \quad \frac{1}{t} = \frac{1}{2e} \frac{F}{(1-F)} \quad \begin{array}{l} d: \text{mean grain size} \\ t: \text{mean twin spacing} \\ e: \text{mean twin thickness} \end{array}$$

Extended Bouaziz-Allain model “TWIP4EU”

Stress-dependent hardening

Stress-dependent twinning evolution:

The factor k_{twin} is introduced to the definition of the twin volume increment :

$$\dot{F} = F_0 m \left[1 - e^{-\beta(s_p - \varepsilon_{\text{init}})} \right]^{m-1} \left[-e^{-\beta(s_p - \varepsilon_{\text{init}})} \right] [-\beta] \dot{s}_p k_{\text{twin}}$$

k_{twin} is a function of the current stress triaxiality η :

$$k_{\text{twin}} = c\eta - \frac{c}{3} + 1, \quad \eta = \sigma_H / \sigma_{\text{eq}}$$

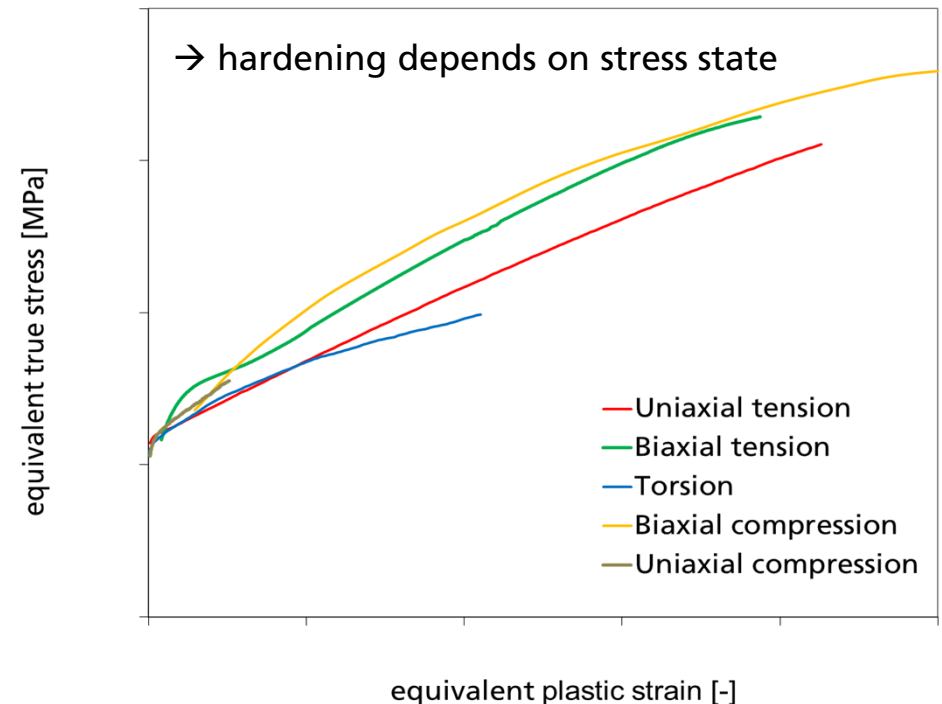
c: slope of a linear function

→ controls the effect of η on twinning

Twin volume evolution becomes

- **Stress dependent**
- **History dependent**

Preliminary analysis from SZMF:



- **Assumption:** evolution of twin volume fraction depends on the current stress state
- Stress dependent twinning was also observed by [Renard et al., 2012, Scripta Mat.]

Extended Bouaziz-Allain model "TWIP4EU"

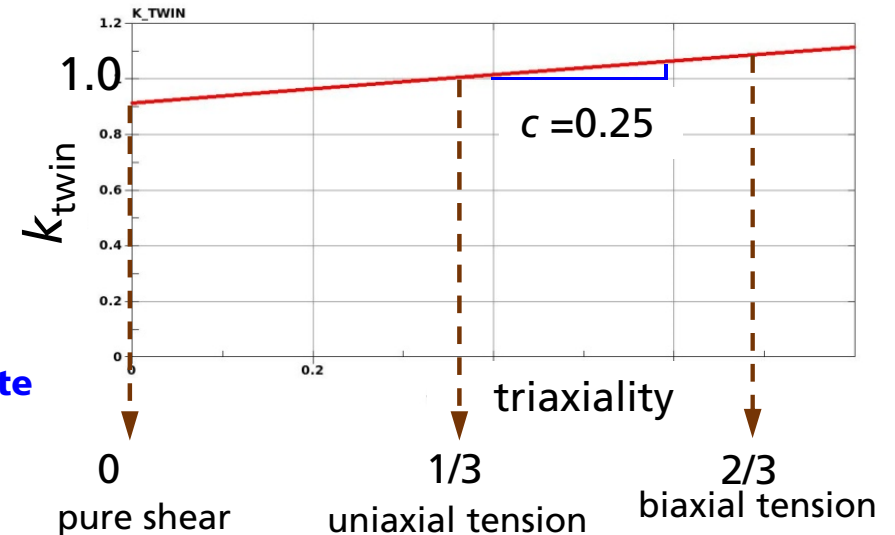
Stress-dependent hardening

- Stress-dependent twinning evolution:

$$\dot{F} = F_0 m \left[1 - e^{-\beta(s_p - \varepsilon_{init})} \right]^{m-1} \left[-e^{-\beta(s_p - \varepsilon_{init})} \right] [-\beta] \dot{s}_p k_{twin}$$

$$k_{twin} = c\eta - \frac{c}{3} + 1$$

$$k_{twin} \left(\eta = \frac{1}{3} \right) = 1 \quad \rightarrow \text{No influence on uniaxial stress state}$$



- Twin volume fraction, F

pure shear

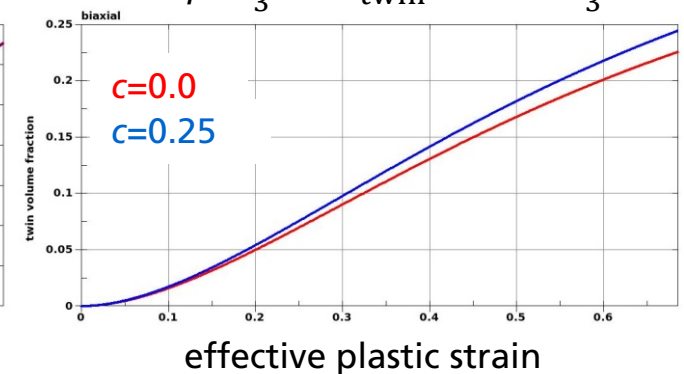
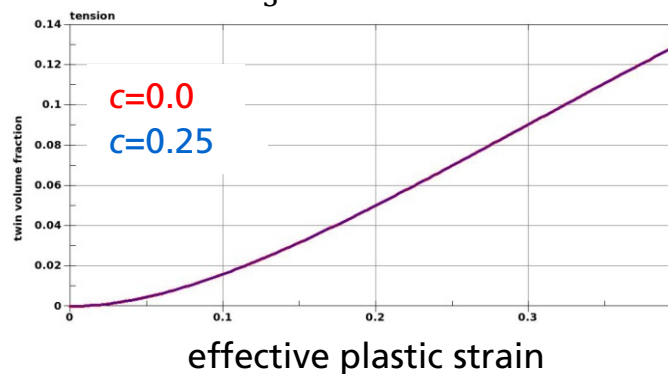
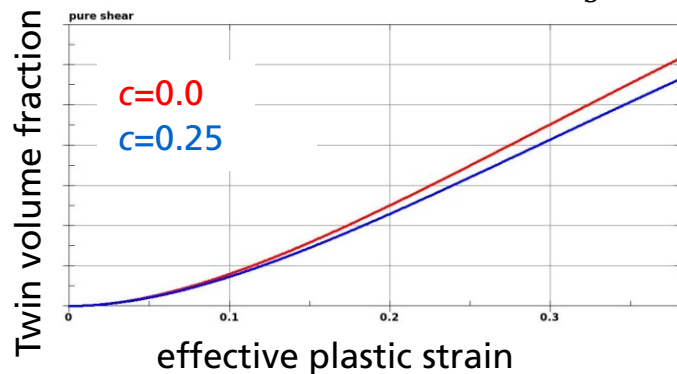
$$\eta = 0.0 \rightarrow k_{twin} = 1.0 - \frac{c}{3}$$

uniaxial tension

$$\eta = \frac{1}{3} \rightarrow k_{twin} = 1.0$$

biaxial tension

$$\eta = \frac{2}{3} \rightarrow k_{twin} = 1.0 + \frac{c}{3}$$



Outline

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- Analysis of the experimental results for TWIP steel
- Macroscopic constitutive model for TWIP steel
- Numerical simulations
 - Model parameter identification
 - Preliminary simulations of metal forming tests
- Conclusions and outlook

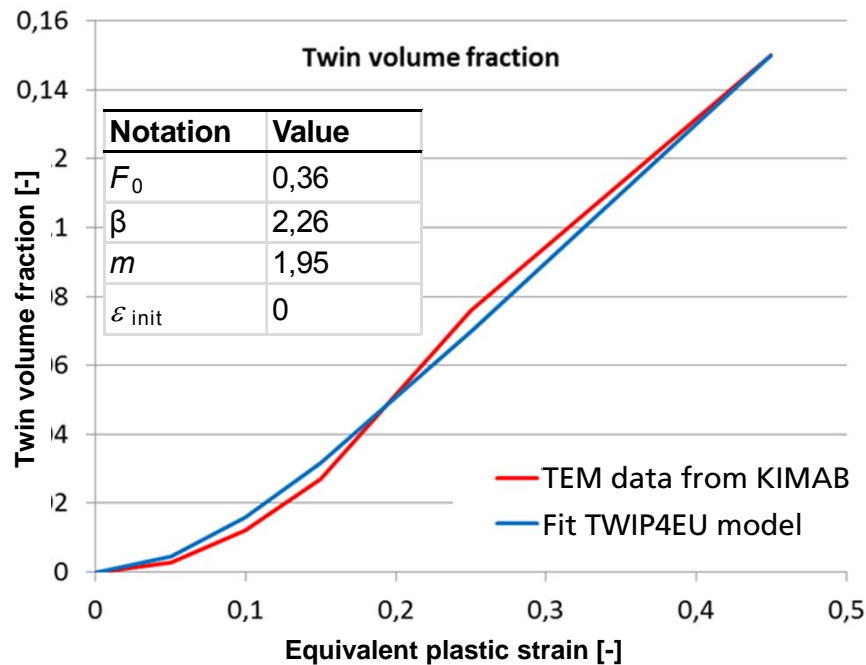
Numerical simulation

Parameter identification strategy

Two-step procedure for parameter identification:

Step 1: Fitting the evolution of the twin volume fraction using the data from microstructure analysis

$$F = F_0 \left(1 - e^{-\beta(s_p - \varepsilon_{init})} \right)^m$$

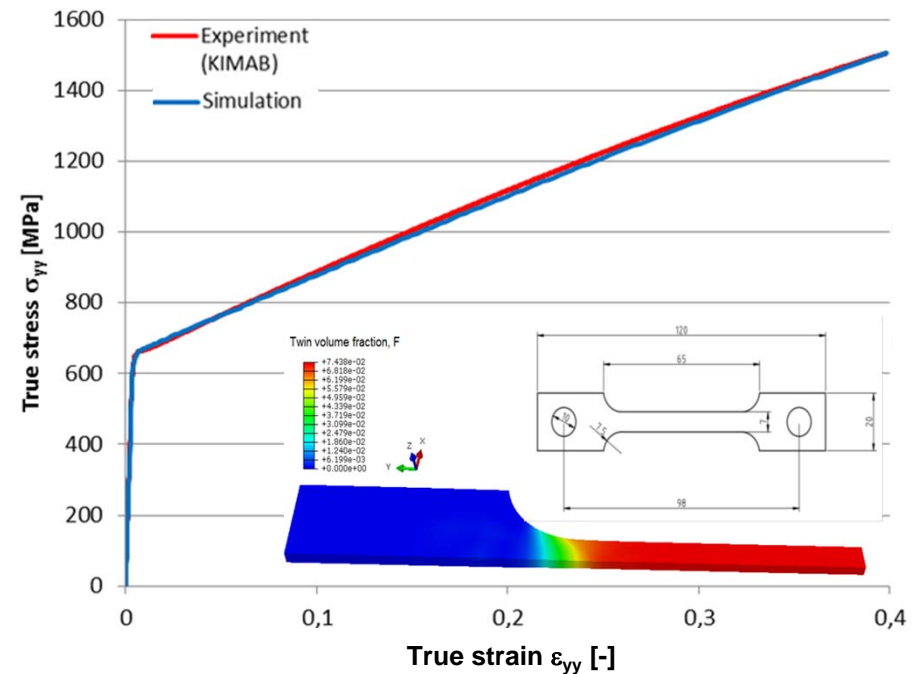


Step 2: Fit the remaining parameters to the uniaxial tensile curve using LS-OPT

$$\alpha = 0,388 \quad \sigma_0 = 590 \text{ MPa}$$

$$k = 0,019 \quad n_0 = 1,5$$

$$f = 1,41$$

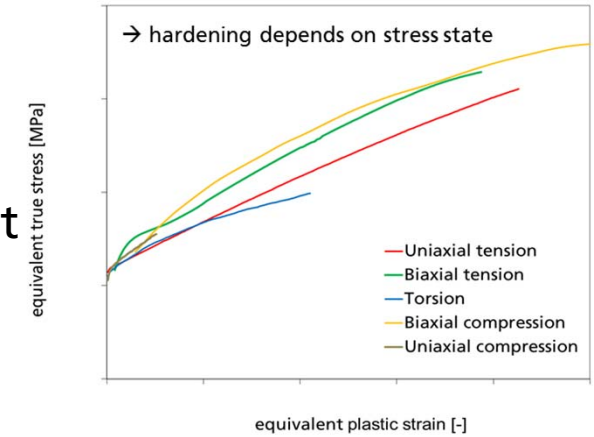


Numerical simulation

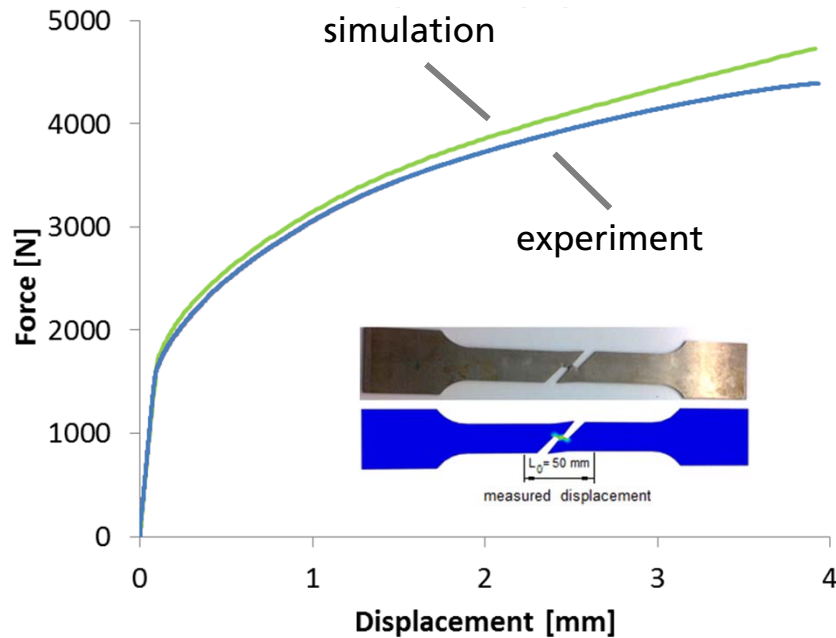
Parameter identification strategy

Shear test:

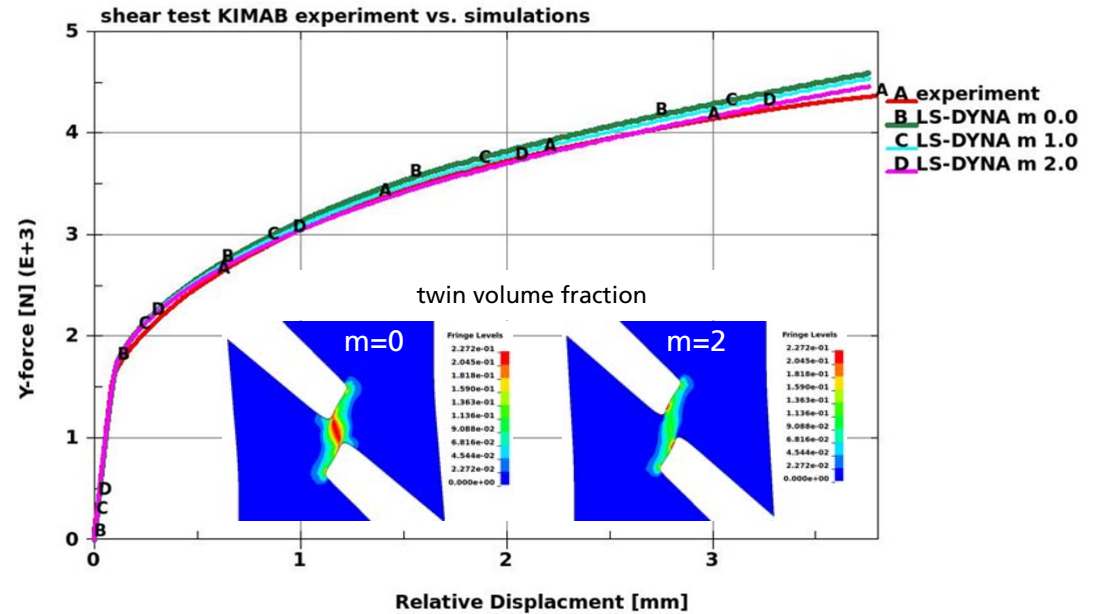
- Some deviations can be observed if the stress dependent hardening is neglected
- Calibration of the shear test using the stress dependent twinning evolution



Shear test without stress dependent twinning



Shear test considering stress dependent twinning → Uniaxial tensile behavior is not influenced

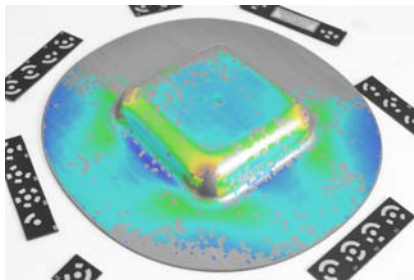


Experimental analysis

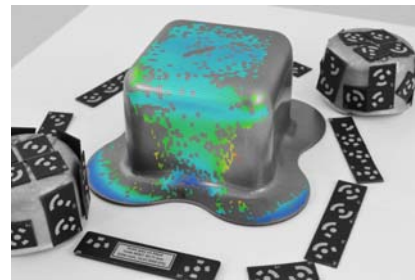
Forming experiments – deep drawing of square cups

Evaluation of strain fields using optical strain analysis tools:

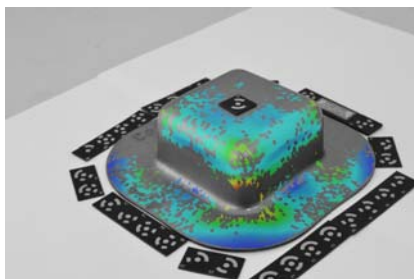
- Deep drawing of square cups
- Very good agreement between evaluated experiments
- High formability of TWIP-steel



draw depth: 20 mm

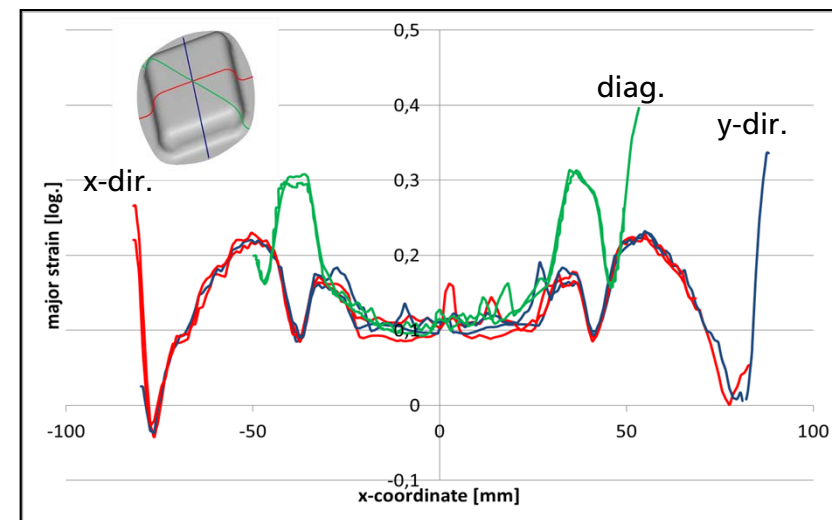
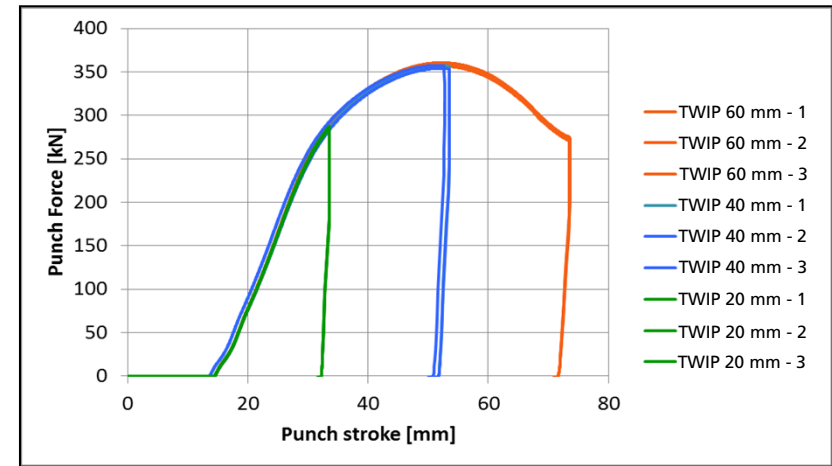


draw depth: 60 mm



draw depth: 40 mm

Strain analysis on square cups with different draw depths. The optical strain analysis tool "Argus" was used for evaluation.



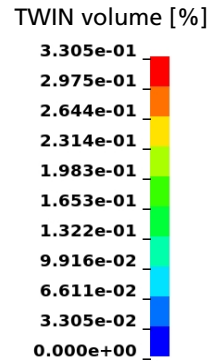
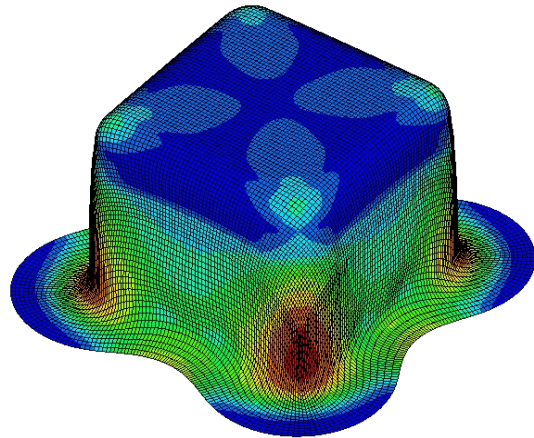
Evaluation of major strain for two square cups, draw depth = 60 mm

Numerical simulation

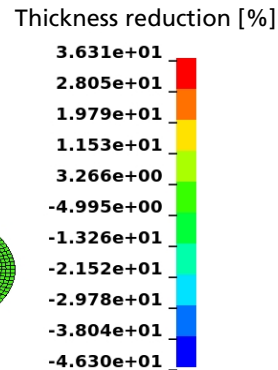
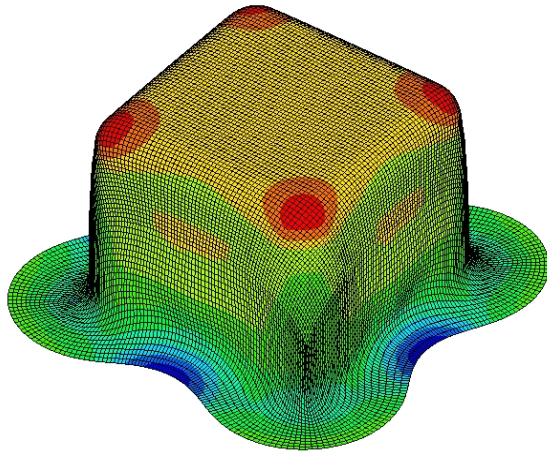
Deep drawing of a square cup

■ Simulation results

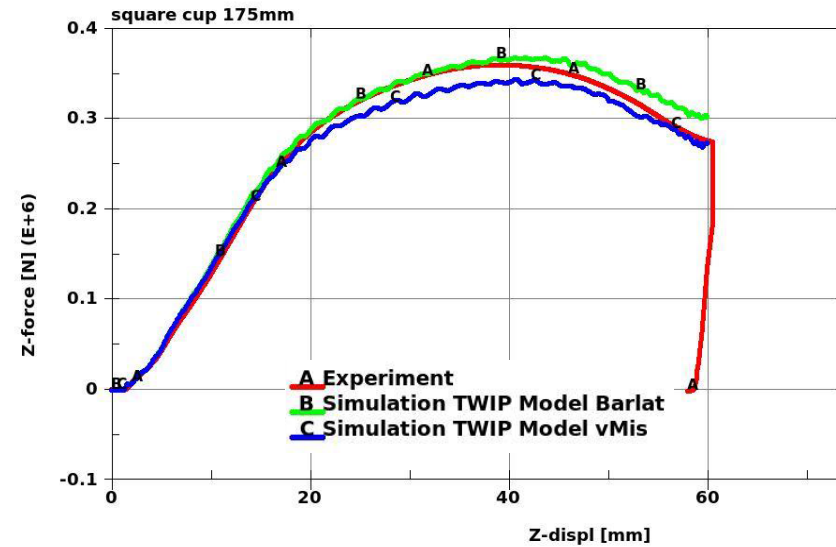
Twin volume fraction



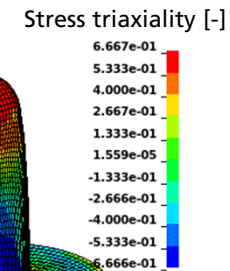
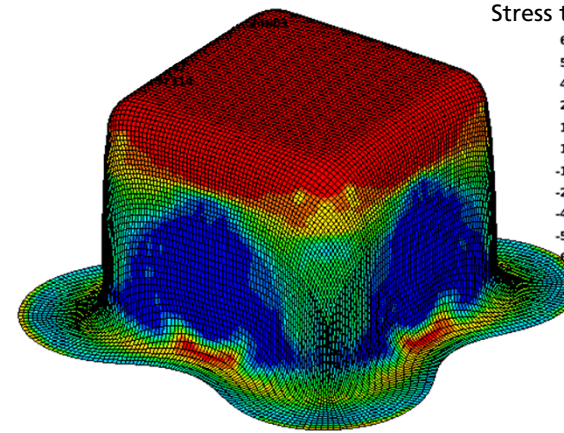
Thickness reduction



Load-Displacement



Stress triaxiality

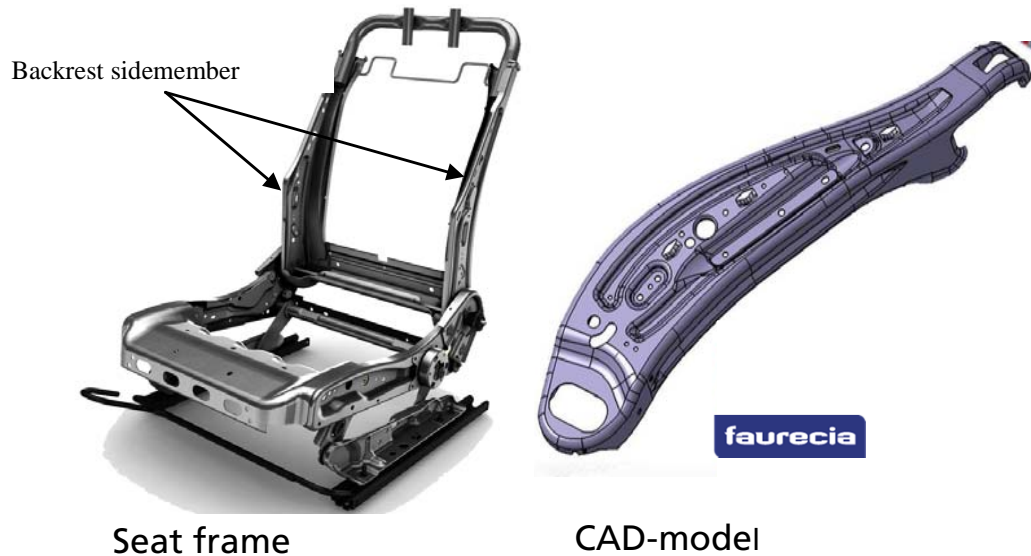


Prototype component

A **backrest sidemember** was chosen as prototype component

- Crash relevant for automotive industry
- Complex 3D-geometry
- Fits to maximum blank dimensions

CAD-Model of the prototype component



TWIP-steel prototype component



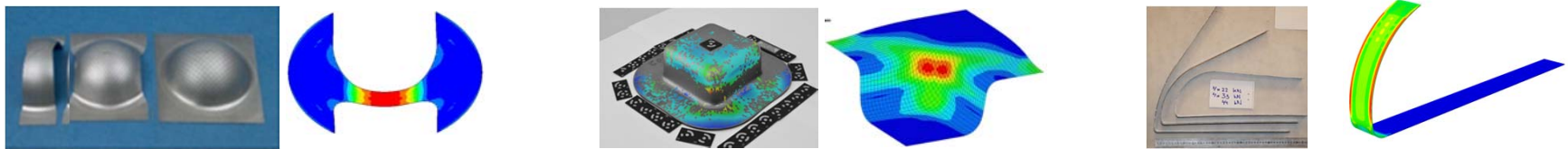
Summary and outlook

Summary

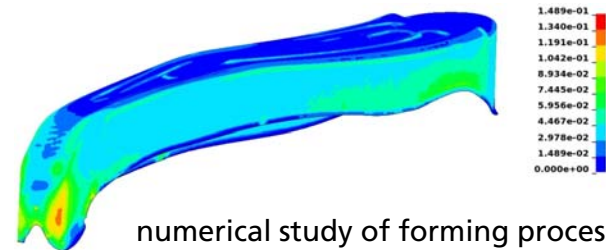
- TWIP-Steel model for sheet metal forming applications
 - Based on a micromechanically motivated approach (Bouaziz)
 - Stress depended twinning
 - Isotrop (v. Mises, 2D & 3D) and anisotropic (Barlat YLD-2000, 2D)
 - Solid and shell formulation available
- First validation of numerical results are in good agreement with experimental data

Outlook

- More accurate description of the Bauschinger-Effect
- Simulation of different forming processes and evaluation of the TWIP-Steel model:



- Simulation of prototype forming process



numerical study of forming process

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Thank you for your attention

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Research Fund for Coal & Steel